THE QUANTITATIVE EVALUATION OF TG-CURVES OF POLYMERS

H.A.SCHNEIDER

Institut fiir Makromolekulare Chemie der Universitat Freiburg,

- Hermann-Staudinger-Haus- Stefan-Meier-Str. 31, D-7800 Freiburg

ABSTRACT

It is shown that any attempt of kinetic interpretation of TG-data of polymer degradation has to consider all peculiarities of nolymer destruction nrocesses. Besides the heterogeneous character of this process, mainly its radical chain reaction nature, the presence of weak bonds and possible scission in chain fragments of different length will prevent stationary state of reaction. Therefore rather a variation of activation energy during non-isothermal degradation of polymers will be the consequence. Supplementary larger polymer chain fragments must not be volatile at temperature of formation.

GENERAL

As everybody knows thermogravimetry (TG) is a method of analysis of mass changes in sample supposed to a well defined temperature programme. The corresponding weight change of the sample will be represented against the temperature and gives the respective TGcurve. The first derivative of this TG-curve is the belonging DTGcurve.

Thermal degradation in condensed phase accompanied by the evolution of volatile reaction products will be a heterogeneous process and will show weight loss during temperature increase. Thermal degradation, however, may occure in several succesive degradation steps.

Taking into account the heterogeneous character of this process the chemical transformation can be exprimed either by the procentual weight change, $(w_m/w_m)x100$, or more convenient by the conversion degree, $\alpha = w_m/w_m$. w_m is the weight loss at the end of the respective degradation step and w_{π} at a given instant at the reaction temperature T.

The degradation process itself my be characterized by the respective temperature range, delimited by the initial, T_i , and final temperature, T_f , of degradation. Sometimes also the temperature of

0040-6031/85/\$03.30 © 1985 Elsevier Science Publishers B.V.

maximum reaction rate, $\texttt{r}_{_{\texttt{M}}}$, is specified. The temperature range of degradation will be shifted to as higher temperatures as higher the heating rate.

Due to the heterogeneous character of the process, diffusion and heat transfer will influence and therefore both the size and the shape of the sample, as well as the surrounding atmosphere and pressure are determinant. Consequently any qualitative comparison of TG-data supposes constant reaction conditions and apparative disposition as well as constant heating rate

Fig.1 Formalized TG/DTG - curves.

As both temperature and time are conversion determining factors in non-isothermal conditions of TG, starting with the simplest law of formal kinetics for the time dependence of the conversion degree, α , (1) and for the temperature dependence of the reaction rate constant, k, (2), it is easy to understand that the steepnes of TG/DTGcurves is rather determined by the activation energY, E, of the process.

 $\frac{d\alpha}{dt}$ = $\kappa f(r\alpha)$; $f(r\alpha)=(1-\alpha)^n$: $F(r\alpha)=\kappa t$ (1) κ = Aexp(-E/RT) : $\frac{d \log \kappa}{d \tau}$ = E $\frac{1}{2.30}$ $2.303 RT²$ (2)

In Fig.1 it is shown that the width of the temperature range is as smaller as higher the activation energy, i.e. the temperature coefficient of the reaction. Therefore it may be hazardous to use other characteristics than T_i to specify the thermal stability of the sample.

Depending on the respective activation energies T_M -temperatures for inctance may coincid, even if the initial temperatures of degradation T_i are quite different as sketched in Fig.1.

QUANTITATIVE ESTIMATION OF TG-DATA

Quantitative estimation of TG-data supposes kinetic interpretation of TG/DTG curves.

Customary it is supposed that the rate equation of heterogeneous processes

$$
d\alpha/dt = k(T) f(\alpha) g(\alpha, T) h(\Sigma x_i)
$$
 (3)

can he used in its simplified form, considering only temperature, $k(T)$, and conversion, $f(\alpha)$, dependent parameters.

$$
d\alpha/dt = k(T) f(\alpha)
$$
 (4)

Conversion-temperature cross terms $g(\alpha,T)$ are assumed ineffective and all physical factors which may influence the heterogeneous process, like sample size and shape, as well as surrounding atmosphere and pressure are eliminated due to the constancy of chosen reaction conditicns.

Taking into account the linear temperature proqramme used in TG

 $hr = dT/dt$ (5) hr being the heating rate in KS **-1 ,** rate equation (4) can be reformulated

 $d\alpha/dT = {k(T)/hr} f(\alpha)$ (6) and than explicited.

Temperature dependence is supposed to obey the Arrhenius law

$$
k(T) = Aexp(-E/RT)
$$
 (7)

whilst conversion function is considered in its simplest form

$$
f(\alpha) = (1-\alpha)^{n}
$$
 (8)

although for heterogeneous solid-gas reaction also more complicated expressions are encountered (ref.1).

In extension of the language of formal gas-phase kinetics, E is considered the activation energy and n the "reaction order" of the heterogeneous process in TG condition.

Eqn.(b) may be now explicited as follows

$$
d\alpha/dT = -\frac{A}{hr} \exp(-E/RT) (1-\alpha)^n
$$
 (9)

which is the basic equation of the non-isothermal kinetics of TG. Separation of variables for integration of the differential

rate equation (9) yields from the beginning two questions.

So the conversion integral supposes for solution the knowledge of the reaction order, n.

$$
F(\alpha) = \int_{0}^{\alpha} \frac{d\alpha}{(1-\alpha)^{n}}
$$
 (10)

As n has no physical meaning or support and therefore cannot be predicted, only trial and error method can be recommended to aproximate n for best fit of experimental data.

Concerning the temperature integral

$$
I(T) = \int_{T_i}^{T} -\frac{A}{hr} \exp(-E/RT) dT
$$
 (11)

no direct solution is accesible and in literature plenty of proposales are encountered for most accurate solution of the temperature inteqrale, including various computer proqrammes. (ref.2)

This explains also the multitude of methods recommended for kinetic interpretation of TG-data, customary classified in differential methods, which starts with the differential rate equation (9) and in integral methods.

Taking into account the aime of this paper, only one of these solutions will be discussed. According to Doyle (ref.3) the fol-Lowing approximate solution can be used if the ratio $E/RT > 15$, condition fullfilled by majority of degradation processes:

$$
\log F(\alpha) = \log(\text{AE/R}) - \log \text{ hr} - 2.315 - 0.457 \frac{E}{R} T^{-1} \qquad (12)
$$

This solution is the startinq point of one of the promising kinetic interpretation methods proposed independently by Flynn and Wall (ref.4) and by Ozawa (ref.5) respectively.

For different heating rates and constant transformation degree, $iso-a$, eqn.(12) can be formulated as follows:

$$
-\frac{d \log \, hr}{d \, T} = 0.457 \, \frac{E}{R} \tag{13}
$$

i.e. a series of parallel isoconversional straight lines will be obtained with slope as steeper as higher the activation energy of the process, as suggested in Fiq.2.

As a consequence the shift of the temperature range of the degradation process with increasing heating rate to higher temperatures will be as more pronounced as smaller the respective activation energy.

KINETICS AND MECHANISM OF POLYMER DEGRADATION

Depending on the nature of the polymer and the structure of the

62

Fig.2 Formalized Flynn-Wall-Ozawa plot of degradation processes with different activation energies

repeating unit during thermal degradation different reaction products result, monomer formation being rather the exception. For vinylic polymers the following general scheme can be presented:

 H -transfer + β -scission

The effective reaction mechanism with side group scission (A') or main chain scission (A" and A"') is related to the polar nature of the X-substituent and the probability of monomer regeneration (A") depends on the absence or the presence of mobil tertiary Hatoms. In the latter case (A"') chain fragments of different length and volatility are formed by inter- or intramoleculare H-transfer followed by R-scission. In principle the chain fragments must not be volatile at temperature of formation.

By side group scission (A') small molecules ar formed in a first degradation step, whilst the main chain is splitting off only in a second step at higher temperatures. This first step can be assimilated with a depolymerization in which "monomer molecules" are formed by consecutive elimination of the side groups "grafted" on the main chain.(ref.6).

Typical TG/DTG curves of representatives of the three different reaction mechanisms are shown in Fig.3. The mechanisms presented are only limiting cases, however, real degradation occuring by more ore less accentuated overlapping of these idealized processes.

Observed slope changes in linear-poly(ethylene), l-PE, degradation may be explained by accumulation of longer chain fragments at lower temperatures and their sudden evaporation at higher temperatures. This effect will be as more pronounced as smaller the heating rate (ref.7).

Polycondensation polymers will show in principle similar degradation mechanism, only no monomer molecules will be formed.

Consequently weight loss measurements in polymer degradation will be in principle effective if degradation occurs accordingly to mechanisms A' or A" when volatile small resp. monomer molecules are formed. On the contrary, if

Fig.3. Degradation in air of 20 mg polymer samples. Hr = 0.21 Ks⁻¹ (Sample characteristics see ref.7) chain fragments of different length are formed (A"'), weight loss measurements may fail. Here molecular weight measurements are quite more effective (see Fig.4). The effectiveness depends supplementary on the kinetic chain length of the degradation process, ν , and the temperature. As higher the temperature as smaller the fragments resulted in process A"'.

Fig. 4 Formalized weight loss and molecular weight change curves in polymer degradation.

Thermal degradation of polymers occurs according to a radical chain mechanism, including initiation (I), propagation (P), propagation by transfer (PT) and termination (T) as shown below.

$$
\mathbf{\hat{R}}_i + \mathbf{\hat{R}}_s \sum_{k,j=1}^{K_{\text{C}} \times \mathbf{P}} \mathbf{P}_i + \mathbf{P}_s^{(=)}
$$

Solution of rate equation assumes stationary state for radical concentration, i.e. eqality between rates of initiation and of termination and depending on the nature of the elementary reactions involved, the kinetic chain length and distribution different solutions of overall reaction rate and activation energy are deduced. $(ref.8)$.

Stationary state in chain reaction occurs allways only after an induction period wich is related with the "life time", τ , of the radicals. Essential featurs of radical accumulation during induction period are exemplified for simple depolymerization process.

Rates of accumulation of radicals and monomer during induction period are given by the expressions:

$$
a\left[\hat{R}\right]/dt = W_i - (V_p + V_t)\left[\hat{R}\right], \quad a\left[M\right]/dt = V_p\left[\hat{R}\right]
$$

Integration for radical concentration yields:

$$
\begin{bmatrix} \mathbf{R} \\ \mathbf{R} \end{bmatrix} = \frac{\mathbf{W}_{i}}{\mathbf{V}_{p} + \mathbf{V}_{t}} \left\{ 1 - \exp \left(-(\mathbf{V}_{p} + \mathbf{V}_{t}) \mathbf{t} \right) \right\}
$$

Using the notations $\tau = 1/(v_p + v_t)$ for the "life time" of the radicals and $v=V_p/(V_p+V_t)$ for the "kinetic chain length" of the chain process one obtains

$$
W = d[M]/dt = v W_i \left\{ 1 - \exp(-t/\tau) \right\}
$$

and

W being the overall reaction rate. It is easy to demonstrate that stationary state supposes $t \gg r$.

Variation of radical concentration, overall reaction rate and of product (monomer) concentration in simple chain reaction is shown in Fig.5 (ref.9).

In above equations V_i are the frequency factors of the respective elementary processes. They are concentration dependent in contrast to the rate constants.

Fig.5. Evolution of simple chain.

In non-isothermal conditions of thermogravimetry, however, stationary state is allways perturbed and only at maximum reaction rate stationary state assumtion is valid. Consequently at the beginning of the degradation process conversion will depend mainly on activation energy of the initiation step, E_i , and than gradual tend the much smaller overall activation energy characteristic of the stationary state. This gradual decrease of activation energy during occurence of process will be observed in any radical chain reaction in non-isothermal condition. As a consequence activation energy will show an apparently conversion dependence, $E=E(\alpha)$.

Formalized Flynn-Wall-Ozawa plot for degradation process with conversion dependent activation energy, typical for thermal degradation of polymers in non-isothermal condition of TG is shown in Fig.6.

Fig.6 Formalized Flynn-Wall-Ozawa plot of degradation process with conversion dependent activation energy.

It is evident that the temperature range is not only shifted to higher temperatures, but also as larger as higher the heating rate. That means that the temperature coefficient of the reaction (i.e. the activation energy) decreases with heating rate if a decrease with conversion degree of activation energy is observed. Vice versa is also valid and will be characteristic of parallel processes in thermogravimetric conditions. Theoretical shapes of TG/DTG curves for parallel processes are analysed by Flynn (ref.10) and Ozawa (ref.11).

It is easy to demonstrate that conversion dependent activation

energy will be allways accompanied by heating rate dependent activation energy.

Starting with the same approximate solution of Doyle - eqn. (12) and assuming at constant heating rate a given conversion range, eqn.(12) can be reformulated as follows: For iso-hr

$$
\Delta \log F(\alpha) = \Delta \log (AE/R) - 0.457 \Delta(\frac{E}{R}T^{-1})
$$
 (14)

For E = const, $\Delta \log(\mathrm{AE}/\mathrm{R})$ =O, whilst for E = E(α) it can be assumed that $\triangle log(AE/R) \ll \triangle \left(\frac{E}{R}T^{-1}\right)$.

Considering now the same conversion range, but at different heating rates, because of the identity

$$
\left[\begin{array}{cc} \Delta & \log F(\alpha) \\ \ln r & \end{array}\right] \quad \text{hr'} \quad \equiv \left[\begin{array}{cc} \Delta & \log F(\alpha) \\ \ln r & \end{array}\right] \quad \text{hr''} \tag{15}
$$

it will be also valid the identity

$$
\left[\Delta \left(ET^{-1}\right)\right]_{hr'} = \left[\Delta \left(ET^{-1}\right)\right]_{hr''}
$$
 (16)

The only condition of validity of eqns. (15) and (16) is that the reaction mechanism remains the same by changing the heating rate. For simple radical chain mechanism this condition is respected.

Conversion independent activation energy Flynn-Wall-Ozawa plot shows parallel iso-conversional straight lines as seen in Fig.2, i.e. $\Delta(T^{-1})$ _{hr'}= $\Delta(T^{-1})$ _{hr"} and therefore also constant activation energy at different heating rates. On the contrary conversion dependent activation energy results in non-parallel straight lines (see Fig.6), i.e. $\Delta(T^{-1})^{\dagger}_{h\mathcal{F}^{\dagger}} \neq \Delta(T^{-1})^{\dagger}_{h\mathcal{F}^{\dagger}}$ and consequently activation energy will show heating rate dependence too, E_{hr} , $\neq E_{hr}$, because of the imposed condition by eqn.(16).

Conversion and heating rate dependent activation energy predicted by Flynn-Wall-Ozawa plot according to Fig.6 is obeserved during non-isothermal TG in air for Poly(vinylchloride) (ref.l2), Poly(vinylacetate) (ref **.73),** Poly(vinylalcoho1) (ref **.14)** as well as for the degradation in nitrogen of α -Cellulose (ref.15), anionic Poly(methylmethacrylate) (ref.16) and Poly(ethyleneterephthalate) (ref.17).

Because of the temperature increase during TG degradation of polymers the reaction mechanism itself may change as well due to the heterogeneous character of the process, when diffusion can

concur the chemical process, but mainly due to the presence of weak bonds. These weak bonds will split preferential at lower temperatures, because the small apport of activation energy. As hiqher the temperature as more reduced the influence of weak bonds and random scission will gain in importance. Parallel reactions will occure in TG condition and theoretical aspects concerning conversion and hr influences on shapes of TG/DTG curves are analysed in literature (ref 11 and 12).

The same effect will be observed in presence of oxygen because of the smaller activation energy of the radical generating process

$$
P_n + O_2 \rightarrow HO_2 + R_n
$$

as compared to the thermal bond splitinq reactions. Taking into account that volatile products will relative rapidly surround the sample special in cruicible, the influence of oxygen will be observed mainly at the beginning of the degradation process of polymers in TG condition.

No further straight lines are obtained in a Flynn-Wall-Ozawa plot as shown in Fig.7 for the degradation of anionic Poly(styrene) in air (ref.18). Even much more sophisticated diaqrammes are obtained for the degradation of linear Poly(ethylene) in air.

Complicated Flynn-Wall-Ozawa diagrammes are therefore indices of complicated degradation mechanismes.

Fig.7 Flynn-Wall-Ozawa-Diagrams of anionic Poly(styrene), M=52000 and of linear Poly(ethylene), SRM 1475 Nat. Bur.Stand., M=18310. TG in air, Sample size 20 mg

Finally some considerations concerning the constancy of the heating rate. Although constancy of heating rate in TG is general supposed taking into account the used linear heating programme, parallel measured DTA curves show that specially in the conversion range due to the reaction heat the heating rate is perturbed in the sample. Small sample size and low heating rate will reduce this influence, however, but in detriment of the accuracy of the measurement. Only DSC type heating will exclud the effect of reaction heat. This type of heating, however, is up to now not applied in TG. Kinetic estimation can consider variable heating rate and general the trend of conversion dependence of activation energy is preserved even if absolute values of activation energy are quite different (ref.7).

In conclusion it can be stated the rigurous quantitative evaluation of TG-data of polymers supposes multiple heating rate measurements, because single heating rate data are dependent on conditions used. By multiple heating rate measurements both the compleity of degradation process and probable reaction mechansim as well as belonging kinetic parameters are accesible.

REFERENCES

- 1. L.G.Harrison in C.H.Bamford and C.F.H.Tipper (Eds.), "Comprehensive Chemical Kinetics", Elsevier, Amsterdam, Vol.II/V
- 2. J.Sestak, V.Savata and W.W.Wendlandt, Thermochim.Acta, 7 (1973) 331-556
- 3. C.D.Doyle, Makromol.Chem., 83 (1964) 220
- 4. J.H.Flynn and L.A.Wall, Polym.Letters, 4 (1966) 323
- 5. T.Ozawa, Bull.Chem.Soc.Japan, 38 (1965)-1881
- 6. I.A.Schneider, J.Polym.Sci., Polym.Symp., <u>64</u> (1978) 641 I.A.Schneider and A.Cs.Biró, J.Therm.Anal., <u>5</u> (1973) 293
- 7. H.A.Schneider, in "Thermal Analysis", Proc. %th ICTA, Bayreuth 1980, Birkhäuser Verlag, Basel, Vol.II., 387
- 8. R.H.Boyd, J.Chem.Phys., 31 (1959) 321; J.Polym.Sci., 49 (1961) 81
- 9. V.N.Kondratiev and E.E.Nikitin, "Kinetics and Mechanisms of Gas Phase Reactions", (in russ.), Nauka, Moscow, 1975, Chap.10
- 10. J.H.Flynn, Thermochim.Acta, 37 (1980) 225
- 11. T.Ozawa, J.Therm.Anal., 9 (1976) 217
- 12. D.Furnica and I.A.Schneider, Makromol.Chem., 108 (1967) 182
- 13. E.Mihai, A.Cs.Bir6, A.Onu and I.A.Schneider, Makromol.Chem., 175 (1974) 3437
- 14. M.Popa, C.Vasile and I.A.Schneider, J.Polym.Sci., Polym.Chem. Ed., **IO** (1972) 3679
- 15. N.Hurduc, I.A.Schneider and Cr.Simionescu, Cel1.Chem.a Techn., 2 (1968) 569
- 16. N.Hurduc and I.A.Schneider, Rev.Roum.Chim., 22 (1977) 945
- 17. C.Birladeanu, C.Vasile and I.A.Schneider, Makromol.Chem., 177 (1976) 121
- 18. I.A.Schneider and N.Hurduc, Makromol.Chem., 178 (1977) 547